



## Original Research

## Oxygen doping enhances piezo-photocatalytic degradation of carbamazepine by molybdenum disulfide

Yanjiao Li<sup>a,b</sup>, Lianfeng Wang<sup>a</sup>, Yuting Ning<sup>c</sup>, Ying Yuan<sup>b</sup>, Xinping Fu<sup>c</sup>, Jun Cui<sup>c,\*</sup>, Yu Jiang<sup>b,\*\*</sup><sup>a</sup> School of Materials Science and Engineering, Dalian Jiaotong University, Dalian, 116028, China<sup>b</sup> State Key Laboratory of Environmental Criteria and Risk Assessment, and Key Laboratory of Groundwater Pollution Simulation and Control of Ministry of Ecology and Environment, Chinese Research Academy of Environmental Sciences, Beijing, 100012, China<sup>c</sup> College of Chemical Engineering, Beijing University of Chemical Technology, Beijing, 100029, China

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## ABSTRACT

Environmental pollution from persistent pharmaceuticals like carbamazepine (CBZ) poses severe risks to aquatic ecosystems and human health, yet conventional treatments struggle with low concentrations and secondary pollution. Piezo-photocatalysis, which harnesses mechanical and solar energies to drive charge separation, offers a promising alternative using materials such as molybdenum disulfide ( $\text{MoS}_2$ ), whose layered structure enables tunable piezoelectricity but is hindered by rapid electron-hole recombination and structural instability. However, the mechanistic role of oxygen doping in repairing sulfur vacancies and enhancing symmetry-breaking for improved performance remains underexplored. Here we show that hydrothermally synthesized oxygen-doped  $\text{MoS}_2$  ( $\text{O}_5\text{-MoS}_2$ ) fully degrades  $2 \text{ mg L}^{-1}$  CBZ in 25 min under combined ultrasound and visible light, achieving a rate constant ( $k_{\text{obs}}$ ) of  $0.13 \text{ min}^{-1}$ —11.4 times higher than undoped  $\text{MoS}_2$ . This stems from oxygen substitution narrowing the bandgap to 1.94 eV, boosting the piezoelectric coefficient to  $63 \text{ p.m. V}^{-1}$  (versus  $26 \text{ p.m. V}^{-1}$ ), and generating a 0.19 V built-in potential that drives charge separation, as confirmed by  $4.18 \mu\text{A cm}^{-2}$  synergistic photocurrents, density functional theory calculations revealing heightened Mo–O charge transfer ( $2.08\text{--}2.36 \text{ e}^-$ ), and finite element simulations of deformation-induced fields. Over five cycles,  $\text{O}_5\text{-MoS}_2$  retains 100% efficiency with minimal Mo leaching (1.9%), reducing product toxicity across fish, daphnid, and algal models. These findings delineate oxygen doping's dual role in defect mitigation and polarization enhancement, paving the way for robust piezo-photocatalytic systems in real-world water purification.

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## 1. Introduction

With the rapid development of global industrialization and urbanization, environmental pollution is becoming increasingly severe [1,2]. Water bodies face a serious, hidden threat from the steady increase in organic pollutants, endangering ecosystems and human well-being [3,4]. Carbamazepine (CBZ), a widely used antiepileptic and analgesic agent, is characterized by low concentrations and difficulty in degrading in the environment [5,6].

Low-concentration CBZ not only causes toxic stress to aquatic organisms but may also trigger health crises such as endocrine disorders and increased drug resistance in humans through the food chain [7]. Traditional biological degradation and physical treatment technologies are plagued by low treatment efficiency, high energy consumption, and secondary pollution when treating low-level CBZ pollution [8–10]. These problems make it difficult to meet actual demands.

Currently, environmental remediation research is focused on piezo-photocatalytic technology [11–13]. Before delving into this technology in depth, it is necessary to briefly review the limitations of photocatalysis and piezocatalysis. On the one hand, although photocatalytic technology can utilize solar energy to drive reactions, it faces the problem of a high recombination rate of

\* Corresponding author.

\*\* Corresponding author.

E-mail addresses: [cuijun@buct.edu.cn](mailto:cuijun@buct.edu.cn) (J. Cui), [jiang.yu@craes.org.cn](mailto:jiang.yu@craes.org.cn) (Y. Jiang).

electron–hole pairs, which restricts catalytic efficiency [14,15]. On the other hand, only a single energy source is typically used, and its ability to handle complex environmental pollutants is insufficient [16]. Piezo-photocatalytic technology innovatively combines mechanical energy and solar energy to drive redox reactions [17], effectively overcoming the bottleneck of the high recombination rate of electron–hole pairs in traditional photocatalysis and significantly enhancing photocatalytic performance [13,18].

Molybdenum disulfide ( $\text{MoS}_2$ ), a typical transition-metal dichalcogenide, offers significant promise for piezo- and photocatalysis due to its layered structure and tunable electronic properties [19,20]. However, pristine  $\text{MoS}_2$  suffers from low catalytic efficiency and insufficient charge separation in practical applications, limiting its large-scale use [21]. To solve this problem, researchers have extensively studied defect engineering and element doping [22,23]. While sulfur vacancies (SVs) improve  $\text{MoS}_2$ 's catalytic performance [19,23], they also reduce its carrier lifetime and structural stability [24–26].

In comparison, oxygen doping can not only repair SVs and enhance material stability but also optimize the piezoelectric response by changing the symmetry of the  $\text{MoS}_2$  structure [27], providing a new direction for improving catalytic performance [28]. Research on low-concentration CBZ degradation via oxygen-doped  $\text{MoS}_2$  in piezo-photocatalysis is preliminary; the links between the electronic structure, charge separation, and catalytic performance remain poorly understood. In addition, traditional heteroatom doping methods suffer from drawbacks such as complex preparation processes and uneven dopant distribution [29–31], which further hinder the in-depth exploration of the piezoelectric photocatalytic performance of oxygen-doped  $\text{MoS}_2$  at the mechanistic level. It is worth emphasizing that the piezoelectric effect is highly significant in photocatalysis. When a piezoelectric material is subjected to pressure, it generates a piezoelectric potential. This potential can create a built-in electric field within the material, thereby promoting the separation of photogenerated carriers (electron–hole pairs) and reducing their recombination probability [17]. This process significantly improves the effective utilization of charges in photocatalytic reactions, thus enhancing catalytic efficiency.

To address prior challenges, we successfully synthesized O- $\text{MoS}_2$  using a simple, efficient hydrothermal method to replace some SVs in sulfur-deficient  $\text{MoS}_2$  with oxygen. This study systematically investigated how CBZ degrades under combined piezoelectricity, photocatalysis, and piezo-photocatalysis. Through a combination of theoretical calculations and experimental verification, we conducted an in-depth analysis of the mechanism by which oxygen doping influences the electronic distribution, charge separation efficiency, and catalytic performance of  $\text{MoS}_2$ . The results of this study are expected to fill the gaps in the literature on the piezo-photocatalytic mechanism of oxygen-doped  $\text{MoS}_2$  and lay a solid theoretical and practical foundation for the development of efficient and stable piezo-photocatalytic materials.

## 2. Materials and methods

### 2.1. Preparation of $\text{MoS}_2$ and O- $\text{MoS}_2$

The synthesis of  $\text{MoS}_2$  and O- $\text{MoS}_2$  catalysts was carried out using a hydrothermal method. In a typical procedure, sodium molybdate dihydrate ( $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$ , 1.2 g) and thiourea ( $\text{CH}_4\text{N}_2\text{S}$ , 0.66 g) were used as starting materials. The reagents were completely dissolved in 60 mL of a mixed ethanol/water solvent (7.6:52.4 v/v). The homogeneous solution was subsequently transferred into a 100 mL Teflon-lined stainless-steel reactor and maintained at 220 °C for 24 h. After the reaction, the resulting O<sub>1</sub>-

$\text{MoS}_2$  material was isolated through centrifugation, followed by sequential washing with ethanol–water mixtures. The purified product was then dried under a vacuum at 60 °C. Samples with different compositions of  $\text{MoS}_2$  and O- $\text{MoS}_2$  were obtained by varying the molar ratios of the precursors, as detailed in the Supplementary Materials (Supplementary Table S1). Please refer to Supplementary Text S1 for a comprehensive list of all chemicals and electrode details.

### 2.2. Characterization of piezo-photocatalysts

The surface morphology of the samples was investigated by a scanning electron microscope (SEM; S-4800, Hitachi, Japan) equipped with an energy-dispersive spectrometer (EDS) and a high-resolution transmission electron microscope (HRTEM; JEM-2100, JEOL, Japan). Subsequently, the sample size distribution was analyzed using GMS 3 software. Crystal structure was characterized by X-ray diffraction (XRD) on an Ultima IV diffractometer (Rigaku, Japan) at a scanning rate of  $2^\circ \text{ min}^{-1}$ . High-resolution X-ray photoelectron spectra (Mo 3 d, S 2p, O 1s, and C 1s) were acquired using an ESCALAB 250Xi spectrometer (Thermo Fisher Scientific, USA). Ultraviolet–visible diffuse reflectance spectra (UV–vis DRS) were collected using an Ultima IV spectrometer (Rigaku, Japan). Piezoelectric properties were evaluated using a piezoresponse force microscope (PFM) using a Dimension Icon atomic force microscope (Bruker, Germany), and the surface potential was detected using Kelvin probe force microscopy (KPFM). Molybdenum concentrations were measured using an inductively coupled plasma mass spectrometry using an instrument from Thermo Fisher Scientific (USA).

Working electrodes were prepared by ultrasonically blending 10 mg of catalyst powder, 50  $\mu\text{L}$  of Nafion, and 5 mL of ethanol in a centrifuge tube. The homogeneous mixture was drop-cast onto indium tin oxide glass and followed by natural air-drying in a laboratory environment for 96 h. All piezo-photoelectrochemical tests, such as chronoamperometry (I–T), electrochemical impedance spectroscopy (EIS), cyclic voltammetry (CV), and Mott–Schottky tests, were carried out in a three-electrode system using a PGSTAT204 electrochemical workstation (Metrohm, Switzerland). For all electrochemical tests, 0.5  $\text{mol L}^{-1}$   $\text{Na}_2\text{SO}_4$  was used as the electrolyte, except for CV measurements to determine the electrochemically active surface area (ECSA), where 0.5  $\text{mol L}^{-1}$   $\text{H}_2\text{SO}_4$  was used. The sonication conditions were based on a BILON-1000Y ultrasonic unit (BILON, China).

### 2.3. Evaluation of degradation performance

To evaluate the catalytic performance, 0.01 g of each catalyst was added to 50 mL of a CBZ solution at 2  $\text{mg L}^{-1}$ . Notably, this concentration is considered low in the field of photocatalytic mechanism research and is relatively close to the concentration of CBZ in the environment [12,32–37]. Subsequently, degradation experiments were conducted under four conditions: in the dark, under visible-light irradiation, with ultrasonic treatment, and under combined ultrasound and visible-light irradiation.

The separation between the light source and the bottom of the reaction container was set at 25 cm. This distance ensures that the reaction solution is fully covered by the light source while minimizing the thermal effect of the heat generated by the light source on the reaction solution through the appropriate spacing. Simulated sunlight was provided by a light source (HSX-F300, NBET, China) equipped with a 350 nm filter. The light source intensity was 200–10000  $\text{mW cm}^{-2}$ , and the light source wavelength ranged from 300 to 2500 nm. The ultrasound machine was provided by a Spring Rain ultrasonic cleaner (China).

The total organic carbon (TOC) values of the solutions obtained after reactions under different reaction conditions were measured using a TOC analyzer (TOC-L CPH, Shimadzu, Japan). CBZ concentrations were quantified by high-performance liquid chromatography (HPLC) using an HPLC system (Agilent, USA) equipped with a C18 column from Phenomenex (California, USA). An ultra-high-performance liquid chromatography tandem mass spectrometry device (Micromass Xevo TQD, Waters, USA) was used to pinpoint major intermediate products. Free radical generation was detected by an electron paramagnetic resonance (EPR) spectrometer with a Bruker Elexsys E560 system (USA). Active-species quenching experiments were conducted to identify the free radicals responsible for the degradation. Using factor analysis, the effects of pH, ultrasonic power, catalyst concentration, pollutant concentration, and pollutant type on the catalysts' degradation performance were analyzed. For specific experimental details, please refer to Supplementary Text S2.

#### 2.4. Theoretical calculations and finite element simulation

We modeled the interactions using DFT with the Generalized Gradient Approximation–Perdew–Burke–Ernzerhof exchange–correlation functional. We applied an on-site Coulomb correction (where  $U$ – $J$ , the difference between the Hubbard Coulomb interaction parameter ( $U$ ) and the Hund's exchange parameter ( $J$ ), is set to 5 eV) to the  $d$  atomic orbital of molybdenum. The plane-wave cutoff energy was set at 259 eV. We employed the standard (blocked Davidson) algorithm with reciprocal-space projection operators and required electron convergence criterion to  $1.0 \times 10^{-5}$  eV. Based on the requirement for a  $k$ -point spacing of  $0.5 \text{ \AA}^{-1}$ , a  $1 \times 1 \times 1$   $k$ -point grid was obtained. The actual  $k$ -point spacing was  $0.46 \times 0.46 \times 0.28 \text{ \AA}^{-1}$ , and we centered the grid at the gamma point. We used a Gaussian smearing with a width of 0.05 eV. Under the force-field condition, an additional external pressure of 1 GPa was applied to the aforementioned parameters.

A single-petal structure was constructed by simulating the O-MoS<sub>2</sub> edge curvature using the “image to curve” function in COMSOL Multiphysics 6.3, based on the single-petal structure [38,39]. We then rotated and assembled these petals to obtain a flower-like O-MoS<sub>2</sub> architecture. The connection point of the single petal was set to zero potential and constrained as a fixed value. To study the effect of internal forces on the piezoelectric strain response, we applied compressive stresses of 1 GPa and 2 GPa along the  $z$ -axis from the upper surface to the lower surface.

### 3. Results and discussion

#### 3.1. Characterization of catalysts

In this study, Na<sub>2</sub>MoO<sub>4</sub>·2H<sub>2</sub>O served as the molybdenum precursor, and ethanol was used as the oxygen source. A series of oxygen-doped MoS<sub>2</sub> materials (labeled O<sub>(1–6)</sub>-MoS<sub>2</sub>) were successfully synthesized using a hydrothermal method. Different oxygen doping levels in MoS<sub>2</sub> were precisely achieved by adjusting the amount of CH<sub>4</sub>N<sub>2</sub>S added (Fig. 1a, Supplementary Table S1). By adjusting the amount of CH<sub>4</sub>N<sub>2</sub>S added, the degree of oxygen substitution for SVs can be regulated, thereby achieving the control of different oxygen doping levels in MoS<sub>2</sub> (Fig. 1a–Supplementary Table S1).

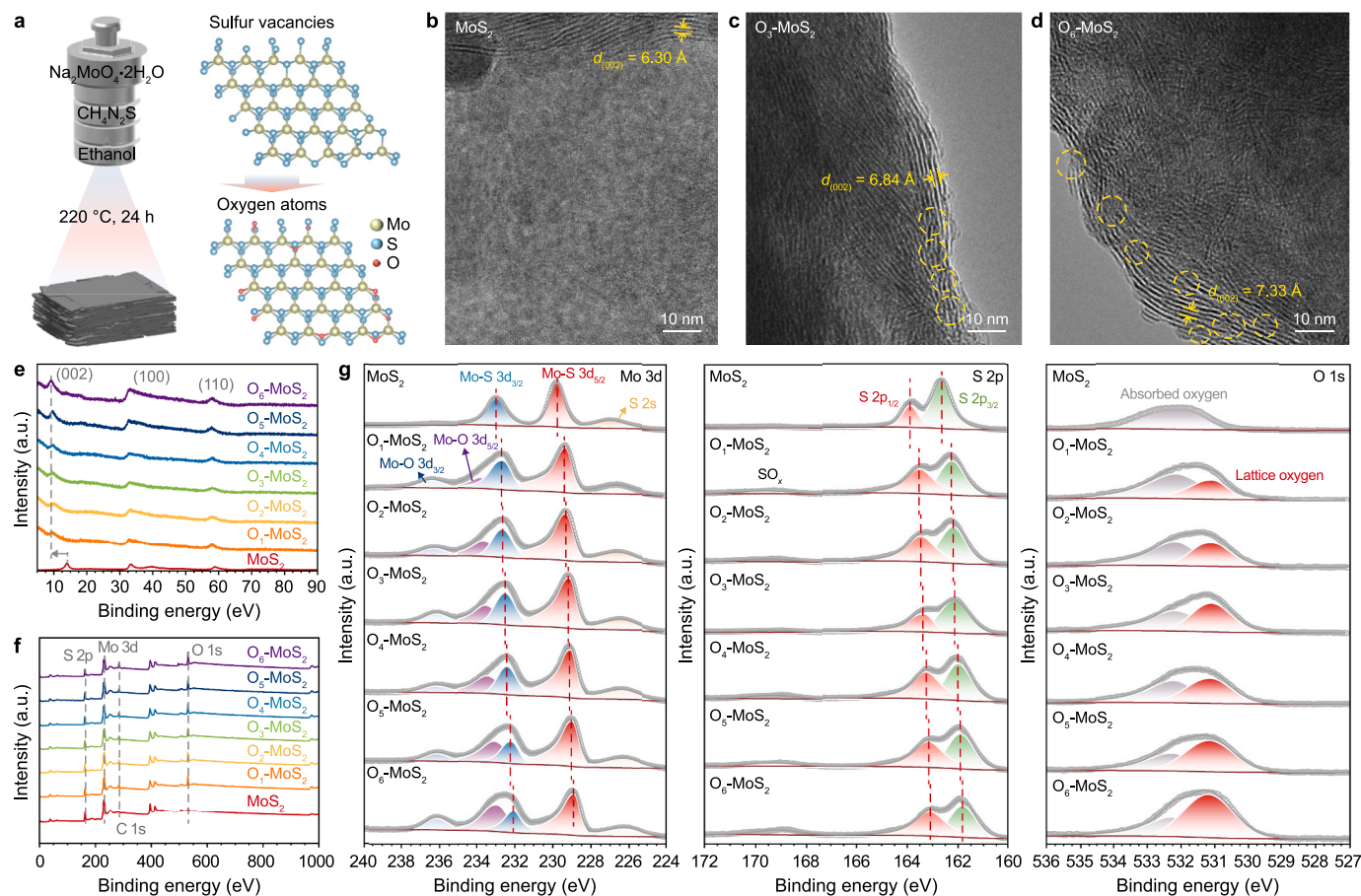
The results of morphological and structural characterization, as shown in SEM images, reveal that all catalysts exhibit flake-like layered MoS<sub>2</sub> structures (Supplementary Fig. S1). Further analysis revealed that the particle size of the O-MoS<sub>2</sub> was positively correlated with sulfur content. The particle diameter gradually decreased from  $2.4 \pm 0.04 \text{ }\mu\text{m}$  of MoS<sub>2</sub> to  $1.6 \pm 0.03 \text{ }\mu\text{m}$  of O<sub>6</sub>-MoS<sub>2</sub>

(Supplementary Fig. S2 and Table S2). The EDS patterns confirmed that the elements Mo, S, and O were uniformly distributed in the O-MoS<sub>2</sub> samples (Supplementary Fig. S3–S9). The HRTEM images indicated that, compared to that of MoS<sub>2</sub>, the structure of O-MoS<sub>2</sub> underwent distortion and featured an increase in active sites (Fig. 1b–d, Supplementary Fig. S10). Moreover, as sulfur content decreased, the O-MoS<sub>2</sub> structure showed a marked increase in disorder. According to crystal structure analysis, the XRD spectra revealed that all samples' diffraction peaks matched those of the hexagonal structure of MoS<sub>2</sub> (PDF#97-003-1067, corresponding to the 2H phase, a thermodynamically stable hexagonal layered crystal structure) (Fig. 1e). Oxygen doping resulted in significant broadening of the (002), (100), and (110) diffraction peaks, as well as shifting their distinct shifting to lower angles. Among these, the (002) peak exhibited the most prominent shift to lower angles [40]. In contrast, the diffraction peaks of (100) and (110) were not significantly affected by the decrease in sulfur content. This phenomenon was attributed to the incorporation of oxygen, which reduces the crystallinity of molybdenum disulfide and disrupts its inversion symmetry [7]. Meanwhile, the introduction of oxygen atoms increases electrostatic repulsion between layers, ultimately widening the interlayer spacing.

The surface electronic states and chemical composition of O-MoS<sub>2</sub> were comprehensively analyzed using X-ray photoelectron spectroscopy (XPS). The Mo 3d peaks in all samples exhibited high similarity, showing comparable chemical states (Fig. 1f). Deconvolution analysis of the high-resolution Mo spectra identified the characteristic peaks of Mo–S bonds at 3d<sub>3/2</sub> (232.7 eV) and 3d<sub>5/2</sub> (229.4 eV), as well as the characteristic peaks of Mo–O bonds at d<sub>3/2</sub> (236.3 eV) and 3d<sub>5/2</sub> (233.8 eV) [41]. The deconvolution analysis of the high-resolution S spectrum identified the characteristic peaks of S(II) at 2p<sub>1/2</sub> (163.6 eV) and 2p<sub>3/2</sub> (162.3 eV). The broad peak observed at 169.5 eV was attributed to SO<sub>4</sub><sup>2−</sup> formed via the partial oxidation of certain sulfur species [42]. In addition, the XPS spectrum (Fig. 1g) showed lattice oxygen at 531.1 eV and adsorbed oxygen at 532.1 eV [43]. To further clarify the quantitative relationship between the oxygen doping level and defect repair, XPS area-integration analysis was performed. The XPS area-integration data indicated that a decrease in the amount of CH<sub>4</sub>N<sub>2</sub>S led to an increase in oxygen doping in O-MoS<sub>2</sub>. Specifically, as the sulfur content decreased, the proportion of peak areas associated with Mo–O bonds and lattice oxygen gradually increased (Supplementary Table S3), which directly reflected the increase in the number of Mo–O bonds. Additionally, the shift of Mo 3d and S 2p peaks to lower binding energies further confirmed the increase in oxygen content in O-MoS<sub>2</sub> [43]. These provided a quantitative correlation between oxygen doping and changes in chemical bonds, which were closely related to SV repair.

In addition, based on XPS atomic percentages, the O-MoS<sub>2</sub> models were constructed using the DFT (Supplementary Table S4) [44,45]. By calculating the activation energy ( $E_a$ ) of oxygen atoms at different positions within the structure, we found that the oxygen atoms in SVs had much lower activation energy than those at S-atom sites or positioned randomly in the lattice (Supplementary Fig. S11). This indicates that the introduced oxygen atoms preferentially occupied the preexisting SV sites and provides a quantitative basis for the defect-repair process from an energy perspective. The results of the DFT energy calculation showed that, compared with undoped MoS<sub>2</sub>, O-MoS<sub>2</sub> had a more stable state, which further explained the effect of oxygen doping from the perspective of the relationship between energy and defect repair (Supplementary Fig. S12). In conclusion, these findings strongly support the successful incorporation of oxygen atoms into MoS<sub>2</sub> and clarify the relationship between oxygen doping level and defect repair.





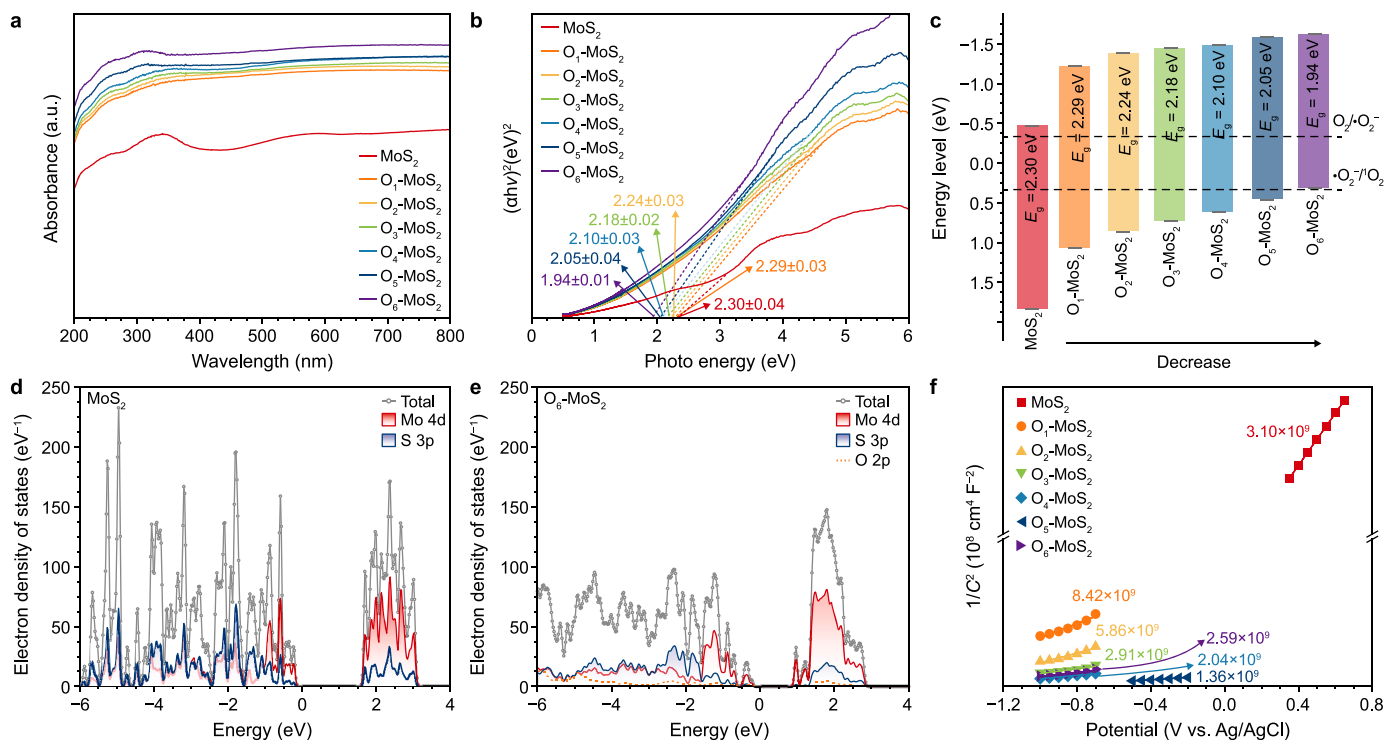
**Fig. 1.** Characterization of O-MoS<sub>2</sub>: Synthesis, morphology, and electronic structure. **a**, Schematic of preparation of O-MoS<sub>2</sub>. **b–d**, High-resolution transmission electron microscope diagram of MoS<sub>2</sub> (**b**), O<sub>3</sub>-MoS<sub>2</sub> (**c**), and O<sub>6</sub>-MoS<sub>2</sub> (**d**). The yellow dashed lines indicate the positions where lattice distortion occurs. **e**, X-ray diffraction patterns. **f**, X-ray photoelectron spectroscopy (XPS) survey spectra. **g**, High-resolution XPS spectra of Mo 3d, S 2p, and O 1s.

The band structure plays a pivotal role in determining the piezoelectric-photocatalytic performance because it directly governs charge separation efficiency, carrier migration behavior, and the generation capability of active species [46]. The optical absorption properties and band structure of the samples were systematically characterized using UV–vis DRS and valence-band X-ray photoelectron spectroscopy (VB-XPS). Compared to MoS<sub>2</sub>, O-MoS<sub>2</sub> demonstrated a substantially extended optical absorption range, indicating that oxygen doping not only preserved the material's ultraviolet absorption capability but also expanded its absorption into the visible light region (Fig. 2a). By applying the Kubelka–Munk equation (Supplementary Text S3 equation (1) [46]), the band gaps ( $E_g$ ) of the different samples were determined to be 2.30 eV (MoS<sub>2</sub>), 2.29 eV (O<sub>1</sub>-MoS<sub>2</sub>), 2.24 eV (O<sub>2</sub>-MoS<sub>2</sub>), 2.18 eV (O<sub>3</sub>-MoS<sub>2</sub>), 2.10 eV (O<sub>4</sub>-MoS<sub>2</sub>), 2.05 eV (O<sub>5</sub>-MoS<sub>2</sub>), and 1.94 eV (O<sub>6</sub>-MoS<sub>2</sub>) (Fig. 2b, Supplementary Fig. S13). The lower  $E_g$ , caused by oxygen doping and the resulting defect states, allowed more electron transitions, leading to significantly improved light absorption. Based on the VB-XPS results and Supplementary Text S3 equation (2) [37], a schematic of the band structure of O-MoS<sub>2</sub> was constructed (Fig. 2c) [36,47]. O-MoS<sub>2</sub> has the potential to generate superoxide radicals ( $\bullet\text{O}_2^-$ ) and singlet oxygen ( $^1\text{O}_2$ ). These active species could further promote the generation of reactive species during the photocatalytic degradation process. According to the partial density of states (PDOS) analysis, the conduction band ( $E_{CB}$ ) and valence band ( $E_{VB}$ ) were predominantly dominated by S 3p and Mo 4d orbitals (Fig. 2d and e, Supplementary Fig. S14).

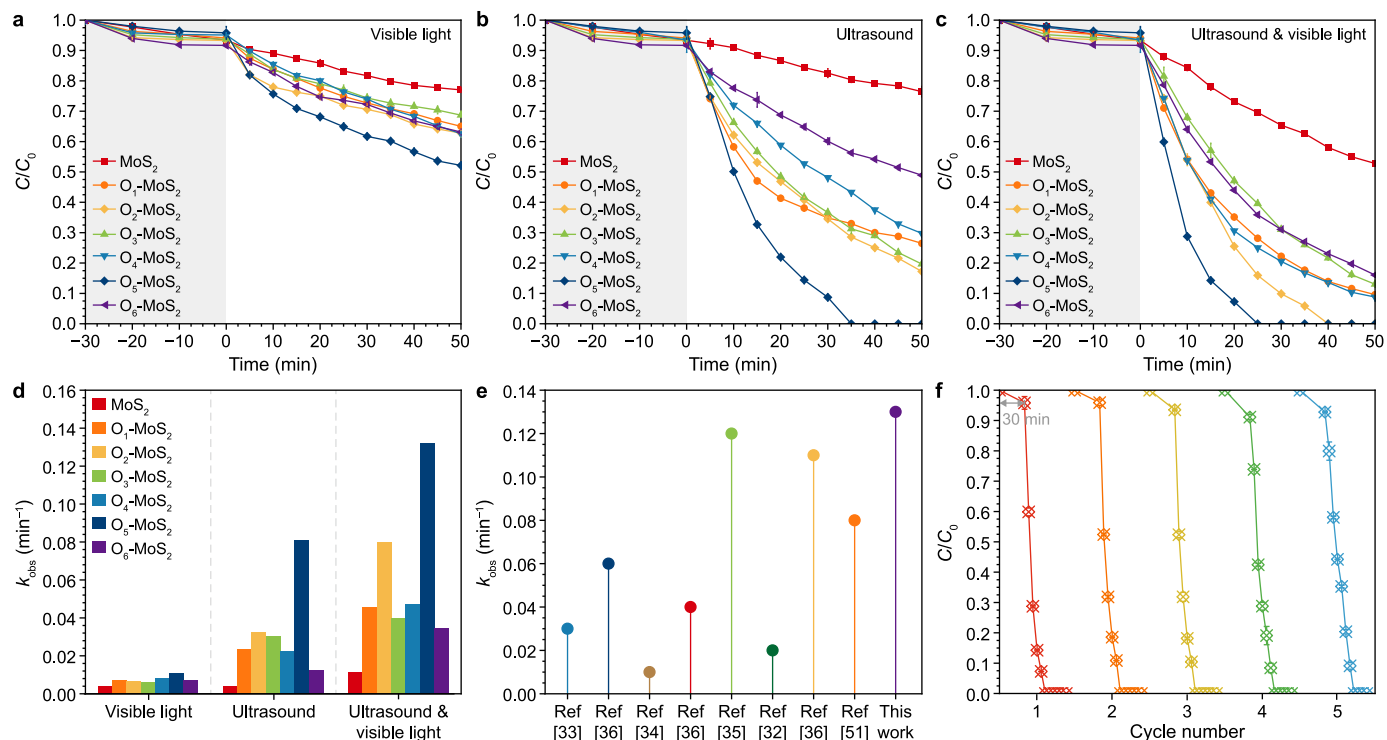
Oxygen doping resulted in a notable negative shift in the  $E_{VB}$ , thereby significantly enhancing light absorption and carrier separation efficiency. Notably, as the proportion of oxygen doping gradually increased, the proportion of O 2p orbitals in the entire system also gradually increased (Supplementary Table S5). This change in oxygen doping significantly affected the material's electronic structure, as evidenced by a distinct negative shift in the valence band. This negative shift effectively enhanced the material's light-absorption capacity and significantly improved the separation efficiency of photogenerated carriers ( $N_d$ ) (Fig. 2f–Supplementary Table S6 and Text S3 equation (3) [48]) [49], further optimizing the photocatalytic performance of MoS<sub>2</sub>.

### 3.2. Piezo-photocatalytic degradation performance

To explore the relationship between photocatalysis and piezocatalysis in O-MoS<sub>2</sub>, the catalytic performance of O-MoS<sub>2</sub> in the degradation of CBZ was analyzed. Experiments were conducted under four conditions: dark, visible light, ultrasound, and a combination of ultrasound and visible light (Fig. 3a–c, Supplementary Fig. S15). In the dark, all samples reached adsorption equilibrium within 90 min, achieving a maximum adsorption rate of 14.2%. Oxygen doping had no significant effect on the adsorption performance of MoS<sub>2</sub> in the dark. Under visible light irradiation, the optimal O-MoS<sub>2</sub> (O<sub>5</sub>-MoS<sub>2</sub>) showed a 24.8% improvement in removal efficiency and a 2.8-fold increase in reaction rate constant ( $k_{obs}$ ) compared to MoS<sub>2</sub> (Fig. 3d) [50]. Notably, the photocatalytic



**Fig. 2.** Optoelectronic and band Structure characterization of MoS<sub>2</sub> and O-MoS<sub>2</sub>. **a**, Ultraviolet-visible diffuse reflectance spectra. **b**, Valence band X-ray photoelectron spectroscopy spectra. Numbers indicate the intersections of the dashed line with the x-axis, corresponding to the free-radical formation potential. **c**, Energy band diagram.  $E_g$ : band gap. **d**, The density of states of MoS<sub>2</sub>. **e**, The density of states of O<sub>6</sub>-MoS<sub>2</sub>. **f**, Mott-Schottky plots. The numbers in the figure represent the slope of the tangent to the Mott-Schottky curves.



**Fig. 3.** Evaluating the efficiency and sustainability of carbamazepine (CBZ) degradation systems. **a–c**, Degradation performance under different conditions: visible light (**a**), 480 W ultrasound (**b**), and visible light and 480 W ultrasound (**c**). **d**, Comparison of reaction rate constants in different systems. **e**, Rate constants for CBZ degradation in the literature and the rate constant in this study. **f**, Stability and reusability tests. The first 30 min (panel **f**; shaded gray regions in panels **a–c**) correspond to the dark adsorption stage.

performance of O<sub>6</sub>-MoS<sub>2</sub> was comparable to that of catalysts with low and medium oxygen content, suggesting that excessive oxygen doping may introduce defects that inhibit photocatalytic activity.

Under ultrasonic conditions, O-MoS<sub>2</sub> catalysts exhibited enhanced piezocatalytic activity compared to undoped MoS<sub>2</sub> (Fig. 3b). The CBZ removal rate increased from 23.6% (MoS<sub>2</sub>) to 100% (O<sub>5</sub>-MoS<sub>2</sub>), and  $k_{obs}$  increased from  $4.2 \times 10^{-3} \text{ min}^{-1}$  (MoS<sub>2</sub>) to  $80.8 \times 10^{-3} \text{ min}^{-1}$  (O<sub>5</sub>-MoS<sub>2</sub>). Among them, O<sub>5</sub>-MoS<sub>2</sub> displayed the highest piezocatalytic activity, followed by O<sub>2</sub>-MoS<sub>2</sub>, indicating that the piezocatalytic activity initially increased, then decreased, and subsequently increased again with increasing oxygen content. Moreover, O<sub>1</sub>-MoS<sub>2</sub>, O<sub>3</sub>-MoS<sub>2</sub>, and O<sub>4</sub>-MoS<sub>2</sub> exhibited similar piezo-catalytic degradation, with O<sub>6</sub>-MoS<sub>2</sub> showing minimal piezoelectric enhancement. This indicates that, before the oxygen-doping amount reaches a certain level, piezocatalytic activity increases substantially. However, once this optimal doping amount is reached, further increases in piezocatalytic activity may be limited by lattice disorder, leading to a decrease.

Notably, the influence of oxygen-doping content on the performance of O-MoS<sub>2</sub> in synergistic catalysis by visible light and ultrasound followed a trend similar to that observed for its impact on pure ultrasonic catalysis. Moreover, under the synergistic effects of visible light and ultrasound, the removal efficiency of CBZ by all materials was significantly higher than that in the individual photocatalytic or piezocatalytic processes (Fig. 3c). Ultrasound significantly enhanced the photocatalytic activity of O-MoS<sub>2</sub>. MoS<sub>2</sub> achieved a CBZ removal rate of 47.2% within 50 min, while O<sub>5</sub>-MoS<sub>2</sub> and O<sub>4</sub>-MoS<sub>2</sub> achieved 100% CBZ removal within 25 min and 40 min, respectively. During the piezo-photocatalytic degradation process, the  $k_{obs}$  of various catalysts showed the following order: O<sub>5</sub>-MoS<sub>2</sub> ( $0.13 \text{ min}^{-1}$ ) > O<sub>2</sub>-MoS<sub>2</sub> ( $0.08 \text{ min}^{-1}$ ) > O<sub>4</sub>-MoS<sub>2</sub> = O<sub>1</sub>-MoS<sub>2</sub> ( $0.05 \text{ min}^{-1}$ ) > O<sub>3</sub>-MoS<sub>2</sub> = O<sub>6</sub>-MoS<sub>2</sub> ( $0.04 \text{ min}^{-1}$ ). Notably, the  $k_{obs}$  value of O<sub>5</sub>-MoS<sub>2</sub> exceeded the relevant values reported in most of the literature (Supplementary Table S7) [32–36,51]. This confirmed the superior piezoelectric-photocatalytic performance of O-MoS<sub>2</sub> (Fig. 3e). Meanwhile, the TOC mineralization rate of O<sub>5</sub>-MoS<sub>2</sub> under the synergistic condition reached 68.8%, which was 3 times and 1.7 times that of individual photocatalysis and individual ultrasonic degradation (Supplementary Fig. S16), respectively, indicating that the synergistic system could achieve efficient mineralization.

Stability tests showed that after five degradation cycles, O<sub>5</sub>-MoS<sub>2</sub> could still achieve complete removal of CBZ within 35 min under the synergistic effect of ultrasound and visible light (Fig. 3f). There was no obvious change in the morphology of the material before and after degradation (Supplementary Fig. S17). The XRD patterns (Supplementary Fig. S18) and XPS spectra (Supplementary Fig. S19) remained consistent, confirming its good stability. In addition, the material exhibited excellent pH adaptability and maintained efficient degradation performance even under low ultrasonic power (Supplementary Fig. S20). Under catalyst dosage of 0.1–0.3 mg L<sup>-1</sup> and pollutant concentration of 1–3 mg L<sup>-1</sup>, the synergistic system achieved 100% degradation of CBZ within 40 min. Extended experiments showed that O<sub>5</sub>-MoS<sub>2</sub> also exhibited good degradation of sulfamethazine, tetracycline, and methylene blue. In the actual surface water system, CBZ degradation could be completed within 35 min under the synergistic condition. The above results confirm that O<sub>5</sub>-MoS<sub>2</sub> has good practical application potential.

### 3.3. Degradation pathway analysis

Two reactive oxygen species ( $\bullet\text{O}_2^-$  and  $^1\text{O}_2$ ) were identified as contributors to the degradation of CBZ through EPR tests (Supplementary Fig. S21). The signals from these active species

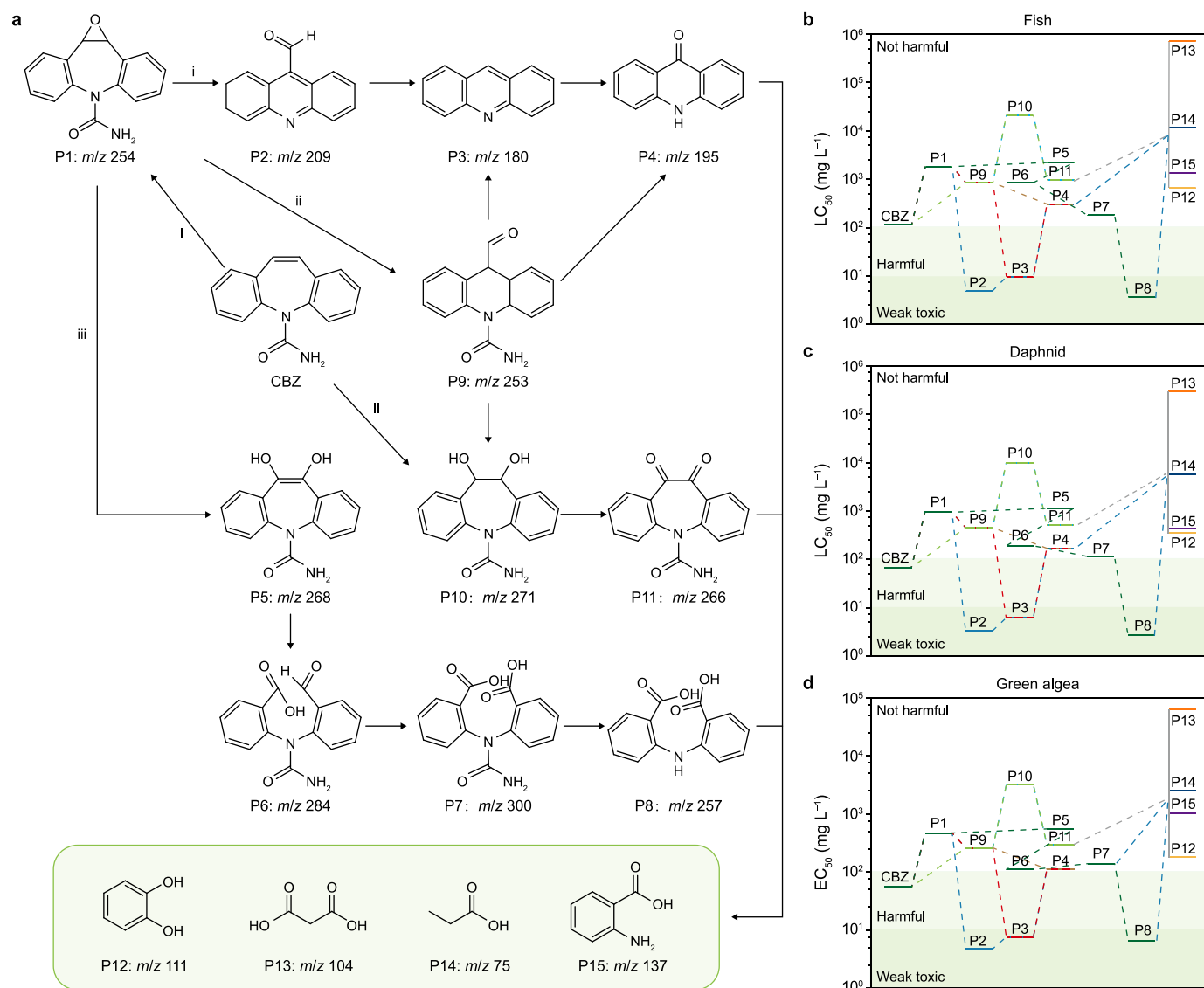
were significantly enhanced when ultrasound was added to the photocatalytic system. This finding indicates that the ultrasound-induced cavitation effect and mechanical vibration effectively enhanced the separation of photogenerated electron-hole pairs and improved the generation efficiency of the active species. In addition, through radical quenching experiments, there was no  $\bullet\text{OH}$  in the system and  $\bullet\text{O}_2^-$  was the main reactive species (56.7%), followed by  $^1\text{O}_2$  (25.8%) and  $\text{h}^+$  (19.2%) (Supplementary Fig. S22).

The intermediate products generated during CBZ degradation were analyzed using a Waters UHPLC-MS system. Sixteen intermediate products were identified (Supplementary Table S8), and two primary degradation pathways for CBZ during the piezo-photocatalytic process were proposed (Fig. 4a) [9,32–34,52,53]. In Pathway I, oxidative ring contraction was identified as the dominant mechanism. CBZ was initially converted to the common intermediate P1. Subsequently, under the influence of reactive species, P1 underwent cleavage and recombination, further generating P2, P5, and P9. In Pathway II, P10 was oxidized from CBZ or P9; additionally, ring cleavage of P7 yielded P11. As the reaction progressed, the intermediate products underwent further deamination and dehydrogenation reactions, ultimately converting large molecules into smaller ones and completing the mineralization process.

The toxicity of CBZ's degradation products was determined using the model of Ecological Structure Activity Relationships (ECOSAR) to understand changes in toxicity during CBZ degradation. As shown in Fig. 4b–d (Supplementary Table S9), the toxicity of the degradation products decreased gradually as the reaction progressed. The results indicate that the piezo-photocatalytic process effectively degraded CBZ and significantly reduced its environmental impact. The molybdenum leaching rate was only about 1.9%. This low leaching level further confirms that the material is unlikely to cause secondary pollution during application and possesses favorable environmental safety and friendliness.

### 3.4. Analysis and discussion of the mechanism

To further elucidate the effect of oxygen doping level on the piezoelectric photocatalytic mechanism of O-MoS<sub>2</sub>, a comprehensive series of photoelectrochemical experiments was conducted to evaluate its photoelectric conversion efficiency. EIS was used to investigate the photogenerated charge transfer behavior at the piezo-photocatalyst/electrolyte interface. In the dark, the charge-transfer resistance ( $R_{ct}$ ) of all samples decreased (Fig. 5a, Supplementary Fig. S23 and Table S10). Oxygen doping effectively promoted charge-transfer kinetics in MoS<sub>2</sub> nanostructures [19]. Under visible-light irradiation, low- and medium-level oxygen doping decreased the  $R_{ct}$  of MoS<sub>2</sub>, with O<sub>5</sub>-MoS<sub>2</sub> reaching a minimum of 6.6  $\Omega$ . However,  $R_{ct}$  increased to 8.2  $\Omega$  in O<sub>6</sub>-MoS<sub>2</sub> due to excessive oxygen doping, which disrupted the active sites and reduced their effectiveness. These findings were confirmed by ECSA measurements conducted under both dark and visible light conditions (Supplementary Fig. S24–S26). Moreover, the transient photocurrent ( $J_{photo}$ ), piezoelectric photocurrent ( $J_{piezo}$ ), and piezo-photocatalytic synergistic photocurrent responses ( $J_{piezo-photo}$ ) of MoS<sub>2</sub> and O-MoS<sub>2</sub> were systematically evaluated to clarify the differences in charge separation efficiency during various catalytic processes [54]. The stabilized  $J_{photo}$  values of all samples were relatively similar, with O<sub>5</sub>-MoS<sub>2</sub> showing only a 0.33  $\mu\text{A cm}^{-2}$  increase compared to MoS<sub>2</sub> (Fig. 5b). The photocurrent density retention efficiency (PDRE) was 60.8% for MoS<sub>2</sub> and 100% for O<sub>5</sub>-MoS<sub>2</sub> (Supplementary Text S3 equation (4) [46]). These results indicate that the oxygen-doping proportion had limited effects on light performance but significantly improved the photostability of MoS<sub>2</sub>. Under ultrasonic conditions, all materials exhibited



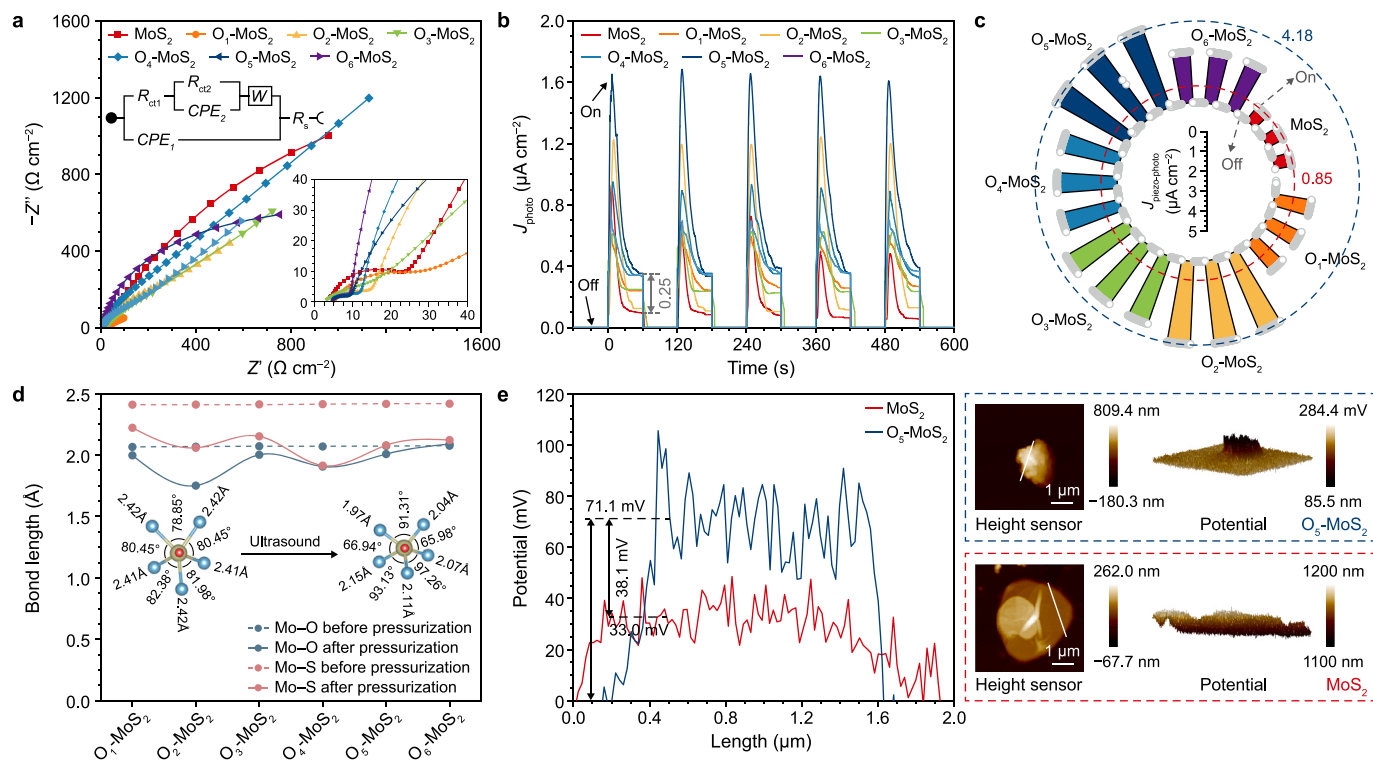
**Fig. 4.** Carbamazepine (CBZ) degradation pathways and ecotoxicity evaluation across multiple organisms. **a**, Possible degradation pathways of CBZ. Roman numerals I and II denote the primary degradation pathways, while lowercase roman numerals i–ii represent the secondary degradation pathways. Degradation products are denoted as PX (X = 1, 2, 3, ...). **b–d**, environmental toxicity assessment of degradation products: fish (**b**), daphnid (**c**), and green algae (**d**). Strong toxic: Half-lethal concentration/Half-effect concentration ( $LC_{50}/EC_{50}$ ) < 1 mg L<sup>-1</sup>, weak toxic: 1 mg L<sup>-1</sup> <  $LC_{50}/EC_{50}$  < 10 mg L<sup>-1</sup>, harmful: 10 mg L<sup>-1</sup> <  $LC_{50}/EC_{50}$  < 100 mg L<sup>-1</sup>, not harmful:  $LC_{50}/EC_{50}$  > 100 mg L<sup>-1</sup>.

excellent instantaneous-response capabilities, reflecting their superior piezoelectric sensitivity. Compared to MoS<sub>2</sub>, the  $J_{piezo}$  of O-MoS<sub>2</sub> was significantly enhanced. Under the synergistic action of ultrasound and visible light, the  $J_{piezo-photo}$  was further increased, surpassing both the individual piezoelectric ( $J_{piezo}$ ) and photo-current ( $J_{photo}$ ) values (Fig. 5c, Supplementary Fig. S27). In descending order of  $J_{piezo-photo}$  values, they were O<sub>5</sub>-MoS<sub>2</sub> (4.18  $\mu A\ cm^{-2}$ ) > O<sub>2</sub>-MoS<sub>2</sub> (4.12  $\mu A\ cm^{-2}$ ) > O<sub>3</sub>-MoS<sub>2</sub> (3.53  $\mu A\ cm^{-2}$ ) > O<sub>4</sub>-MoS<sub>2</sub> (3  $\mu A\ cm^{-2}$ ) > O<sub>6</sub>-MoS<sub>2</sub> (2.53  $\mu A\ cm^{-2}$ ) > O<sub>1</sub>-MoS<sub>2</sub> (1.78  $\mu A\ cm^{-2}$ ) > MoS<sub>2</sub> (0.85  $\mu A\ cm^{-2}$ ). Moderate and high proportions of oxygen doping significantly enhanced catalytic performance under ultrasonic conditions. However, when the proportion of oxygen doping was too high, the activity of O-MoS<sub>2</sub> decreased instead. These observations confirm the degradation performance discussed previously. Meanwhile,  $J_{piezo}$  and  $J_{piezo-photo}$  values showed a strong linear relationship with ultrasonic power across different ultrasonic power conditions (Supplementary Fig. S28). This indicates that ultrasound

significantly enhanced the charge-separation efficiency and carrier mobility in photocatalysis.

The piezoelectric property of the material originates from its noncentrosymmetric crystal structure and excellent deformation ability under external force [55]. To thoroughly analyze this property, this research calculated the Bader charges, bond lengths, and bond angles of MoS<sub>2</sub> and O-MoS<sub>2</sub> at atmospheric pressure and 1 GPa pressure based on DFT [56]. The Bader charge value of the Mo–O bond (2.08–2.36 e<sup>-</sup>) was found to be significantly higher than that of the Mo–S bond (1.59–1.94 e<sup>-</sup>) (Fig. 5d, Supplementary Fig. S29–S33 and Table S11), suggesting Mo–O's higher reactivity. Meanwhile, both the Mo–S and Mo–O bonds showed clear stretching. The stretching amplitude of the Mo–S bond was significantly greater than that of the Mo–O bond (Supplementary Table S12). Under pressure, the bond-angle parameters also changed accordingly (Supplementary Table S13). The above results indicate that O-MoS<sub>2</sub> exhibited excellent structural deformation ability and polarization characteristics under external force,





**Fig. 5.** Electrochemical response and structural alterations of MoS<sub>2</sub> and O-MoS<sub>2</sub> under visible light, ultrasound, and pressure. **a**, Electrochemical impedance spectroscopy (EIS) curves under visible light irradiation. The inset is an enlarged view of the EIS curves. **b**, Transient current density-time curves under on-off cycles of visible light. The stable  $J_{\text{photo}}$  difference between MoS<sub>2</sub> and O<sub>5</sub>-MoS<sub>2</sub> is 0.25  $\mu\text{A cm}^{-2}$ . **c**, Transient current density-time curves under on-off cycles of visible light and 480 W ultrasound. **d**, The charge values of Mo, S, and O under ambient pressure and 1 GPa pressure. The inset figure is the schematic diagram of Mo-O and Mo-S bond lengths and S-Mo-S bond angles. **e**, Surface potential curves. The numbers marked in the left panel represent the average potential values. The right panels are topographic images and Kelvin probe force microscopy potential images for MoS<sub>2</sub> and O<sub>5</sub>-MoS<sub>2</sub>.

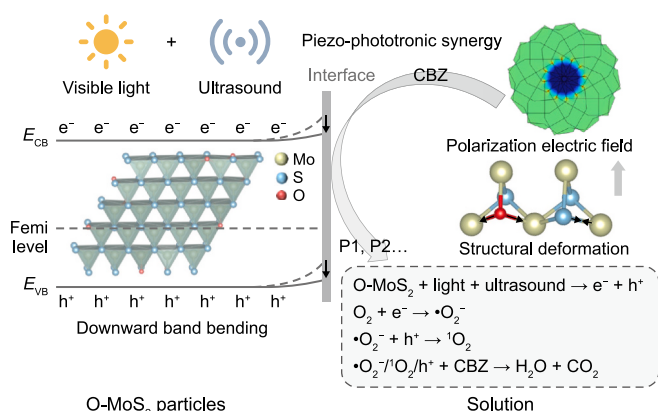
thereby providing a theoretical basis at the structural level for its enhanced piezoelectric performance. Further analysis revealed that mechanical deformation induced by an external force redistributed the Bader charges of Mo, S, and O atoms in O-MoS<sub>2</sub>, thereby forming an internal electric field, thereby revealing the essential mechanism of its piezoelectric effect. COMSOL Multiphysics was used to evaluate deformation-induced piezoelectric activity. The O-MoS<sub>2</sub> model was constructed based on the morphology observed in SEM images and the physicochemical parameters of MoS<sub>2</sub>. Under stimuli of 1 GPa and 2 GPa, substantial deformation and potential intensity were observed (Supplementary Fig. S34) [38,39]. These results indicate that O-MoS<sub>2</sub> nanosheets exhibit robust piezoelectric activity.

To evaluate the enhancement effect of piezoelectric properties on charge separation efficiency, this study systematically characterized parameters related to the piezoelectric field. In PFM measurements, both MoS<sub>2</sub> and O<sub>5</sub>-MoS<sub>2</sub> demonstrated typical butterfly-shaped amplitude loops and hysteresis curves under an applied voltage of  $\pm 10$  V (Supplementary Fig. S35), further confirming their piezoelectric characteristics [57,58]. Additionally, the  $d_{33}$  value of O<sub>5</sub>-MoS<sub>2</sub> (63 p.m. V<sup>-1</sup>) was substantially higher than that of MoS<sub>2</sub> (26 p.m. V<sup>-1</sup>) (Supplementary Fig. S36), highlighting the superior piezoelectric performance of O<sub>5</sub>-MoS<sub>2</sub>. KPFM measurements provided a clear visualization of the surface potential distribution. The results indicate that the surface potential of O<sub>5</sub>-MoS<sub>2</sub> exceeded that of MoS<sub>2</sub> by 38.1 mV (Fig. 5e), reflecting a stronger built-in electric field in O<sub>5</sub>-MoS<sub>2</sub>. This enhanced electric field provided a stronger driving force for accelerating charge-carrier separation [58]. The built-in potential, which drives the directional migration of photogenerated charges, was significantly

enhanced under ultrasonic-visible light conditions for both MoS<sub>2</sub> and O-MoS<sub>2</sub>. Notably, O<sub>5</sub>-MoS<sub>2</sub> (0.19 V) exhibited a higher built-in potential intensity than MoS<sub>2</sub> (0.13 V) (Supplementary Fig. S37). Further research revealed that the magnitude order of the built-in potential of O-MoS<sub>2</sub> with different oxygen doping amounts was extremely similar to the distribution of  $J_{\text{piezo-photo}}$  mentioned earlier, suggesting that oxygen doping effectively enhanced the charge driving force in O-MoS<sub>2</sub>.

Together, these results support the following piezo-photocatalytic mechanism of O-MoS<sub>2</sub> (Fig. 6). The Fermi level of O-MoS<sub>2</sub> was lower than that of the CBZ solution (Supplementary Table S14), which led to the formation of a positive band-bending structure in the space-charge region on the catalyst surface [59]. Under visible-light irradiation, O-MoS<sub>2</sub> absorbed photon energy to generate electron-hole pairs, which tend to recombine easily without external assistance. After the introduction of ultrasound, due to O-MoS<sub>2</sub>'s oncentrosymmetric crystal structure, mechanical vibration caused the positive and negative charge centers to shift, generating an internal electric field that further bent the band downward. This not only lowered the energy barrier but also caused electrons and holes to move in opposite directions, reducing recombination and improving utilization efficiency. The separated electrons reacted with adsorbed O<sub>2</sub> to produce  $\bullet\text{O}_2^-$ , and  $\text{h}^+$  generated  $^1\text{O}_2$ , which promoted the degradation of CBZ. The synergy between piezoelectricity and light stems from the energy provided by light that generates charge carriers and the improved charge separation resulting from the internal electric field induced by ultrasound through the piezoelectric effect, which greatly improves interfacial charge-separation efficiency, intensifies the photocatalytic process, enables more holes and electrons to





**Fig. 6.** Schematic illustration of piezoelectrically enhanced photocatalysis of O-MoS<sub>2</sub>. CBZ: carbamazepine. Degradation products are denoted as PX (X = 1, 2, 3, ...).  $E_{CB}$ : the conduction band.  $E_{VB}$ : the valence band.

participate in the reaction, and significantly enhances the piezo-photocatalytic degradation performance. In addition, the Mo–S bond was a reactive chemical bond. Under visible-light irradiation, as the proportion of oxygen doping in O-MoS<sub>2</sub> gradually increased, the number of reactive bonds increased. Meanwhile, the band gap of O-MoS<sub>2</sub> decreased gradually, and its photoactivity increased accordingly. However, when the oxygen-doping proportion exceeded a specific threshold, the degree of lattice disorder increased significantly, negatively affecting photocatalytic activity. Compared with the performance under pure photocatalytic conditions, the proportion of oxygen doping had a more significant impact on ultrasonic catalytic performance. The driving force of the built-in potential enhanced the current density and the separation efficiency of electron–hole pairs, resulting in a variation trend of the piezoelectric catalytic performance of O-MoS<sub>2</sub> with the increase of oxygen doping ratio: first increasing, then decreasing, rising again to reach a maximum, and finally decreasing. Moreover, the impact of the oxygen-doping proportion on the synergistic catalytic effect was similar to that on the ultrasonic catalytic performance.

## 4. Conclusions

In this study, the piezo-photocatalytic performance of MoS<sub>2</sub> was successfully optimized via an oxygen-doping strategy. Oxygen doping at low, medium, and high levels enhanced the charge-separation efficiency and carrier concentration of MoS<sub>2</sub>. DFT calculations, photoelectrochemical experiments, and COMSOL Multiphysics simulations indicated that oxygen doping at these levels increased the charge difference between Mo and S (from 1.77 e to 1.86 e), thereby enhancing catalytic activity. Under piezoelectric conditions, O-MoS<sub>2</sub> with high-level oxygen doping demonstrated excellent deformation ability and polarization characteristics under external force, along with an increase in the built-in potential (from 0.12 V to 0.19 V). This significantly improved the piezoelectric response and charge-separation efficiency of MoS<sub>2</sub>. O<sub>5</sub>-MoS<sub>2</sub> completed the degradation of CBZ within 25 min, and its  $k_{obs}$  was 11.4 times that of unmodified MoS<sub>2</sub>. This study provides new insights into the impact of oxygen doping on the catalytic performance of MoS<sub>2</sub>, and provides a theoretical basis and practical guidance for designing highly efficient, stable piezo-photocatalytic materials.

## CRediT authorship contribution statement

**Yanjiao Li:** Writing - Original Draft, Visualization, Investigation, Formal Analysis, Data Curation, Conceptualization. **Lianfeng Wang:** Writing - Review & Editing, Methodology, Conceptualization. **Yuting Ning:** Writing - Review & Editing, Visualization, Formal Analysis. **Ying Yuan:** Resources, Conceptualization. **Xinping Fu:** Visualization, Formal analysis. **Jun Cui:** Writing - Review & Editing, Resources, Methodology, Conceptualization. **Yu Jiang:** Writing - Review & Editing, Resources, Investigation, Conceptualization.

## Declaration of interest statement

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The author is an Editorial Board Member/Editor-in-Chief/Associate Editor/Guest Editor for this journal and was not involved in the editorial review or the decision to publish this article.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

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## Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.ese.2025.100639>.

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